Evaluation of Born and local effective charges in unoriented materials from vibrational spectra

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We present an application of the Lorentz model in which fits to vibrational spectra or a Kramers Kronig analysis are employed along with several useful formalisms to quantify microscopic charge in unoriented (powdered) materials. The conditions under which these techniques can be employed are discussed, and we analyze the vibrational response of a layered transition-metal dichalcogenide and its nanoscale analog to illustrate the utility of this approach.

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I. INTRODUCTION

Born and local (or ionic) effective charges are well-known quantities with which to assess chemical bonding and polarization in a material. $\frac{1}{2}$ $\frac{1}{2}$ $\frac{1}{2}$ A large local effective charge indicates a highly ionic system (for instance, 0.8 in NaCl), a medium-sized value points toward an intermediate bonding case (0.4 in GaAs), and a small value is associated with a covalent material $(0 \text{ in } Si).^{3,4}$ On the other hand, Born effective charge describes the static and dynamic polarizations.⁵ From the optical properties point of view, large longitudinaloptic-transverse-optic (LO-TO) splittings are well-known characteristics of polarizable compounds and associated with substantial Born effective charges. This is because LO-TO splitting is directly proportional to charge within Born's original formalism[.1](#page-6-0) Using the LO-TO splitting and high frequency dielectric constant, it is relatively straightforward to quantify chemical bonding from optical measurements of high-quality single-crystal samples and compare the extracted value(s) with first-principles calculations. $6-8$ $6-8$ Unfortunately, there are many instances when single crystals of a bulk material are unavailable, either because they cannot be grown or are not of sufficient size or quality for optical measurements. At the same time, nanoparticles, nanotubes, and alloys (or composite mixtures) present scientifically compelling problems, $9-15$ $9-15$ where optical measurements on powdered materials are the only option, a drawback that complicates the situation but does not diminish the desirability of obtaining quantitative Born and local effective charge data.

Following Born and Szigeti, $\frac{1}{2}$ we present an application of the Lorentz model in which fits to vibrational spectra or a Kramers Kronig analysis are employed along with several useful formalisms to quantify microscopic charge in unoriented (powdered) materials, assuming that the effects of ionic displacement and atomic polarizability are superimposable. We demonstrate that this technique can be used to assess chemical bonding and local strain under certain conditions, a development that advances the field of nanoscience and, at the same time, retains many attractive features of optical spectroscopy and the traditional Lorentz model. This paper is organized as follows. Sections [II](#page-0-0) and [III](#page-2-0) review the situation for single and multiple collinear oscillators. Sections [IV](#page-2-1) and [V](#page-2-2) present how this approach must be modified for a randomly oriented (powder) sample. Section [VI](#page-3-0) illustrates the use of this technique to quantify Born and local (or ionic) charge in powdered $2H-MoS_2$ and its nanoscale analog. Comparison of our analysis of the powdered $2H-MoS₂$ data with that of the single crystal shows that this approach accounts almost perfectly for sample orientation[.16](#page-7-3)[–18](#page-7-4) The extension to assess size effects in nanoparticle samples demonstrates its utility[.16](#page-7-3) Our objective is to clearly present the framework, useful equations, and an application of this method. The Appendix provides alternate frameworks of the model.

II. COLLINEAR OSCILLATOR

A. Review of the general formalism

Starting with the Lorentz model for bound charge, 5.19 one has

$$
m\ddot{x} + m\gamma \dot{x} + m\omega_0^2 x = qE,\tag{1}
$$

where x is the relative displacement of positive and negative ions, *q* is the ionic effective charge, *m* is the reduced mass, *E* is the electric field, γ is the damping parameter, and ω_0 is the oscillator frequency. This model is applicable to vibrational modes of isolated oscillators such as those in gas-phase molecules. Due to dipole-dipole interactions between oscillators in a solid so-called depolarization effects must be considered. Here, *E* has to be replaced by the microscopic field *Emic*, [20](#page-7-6)

$$
E \to E_{mic} = E + \eta \frac{P}{\epsilon_0},\tag{2}
$$

where E_{mic} is the microscopic field that is really acting on the oscillator, *E* is the macroscopic field that should be used to calculate the dielectric constant, P is the polarization, and η is the depolarization factor which depends on the topological arrangement of the oscillators. With this substitution, Eq. ([1](#page-0-1)) becomes

$$
m\ddot{x} + m\gamma \dot{x} + m\omega_0^2 x = q\left(E + \eta \frac{P}{\epsilon_0}\right).
$$
 (3)

The overall goal is to find the dielectric constant which relates E and P . However, in Eq. (3) (3) (3) , there are three unknown variables: *x*, *E*, and *P*. One more equation is needed to find the relationship between *E* and *P*. This additional information will come from the definition of *P*. Assuming the polarization *P* is a linear combination of ionic contributions $P_i = \frac{xq}{V}$, which comes from relative displacements of ions, and electronic contributions $P_e = \frac{\epsilon_0 \alpha}{V} E_{mic}$, which is due to the distortion of electron clouds, one has

$$
P = P_i + P_e = \frac{xq}{V} + \frac{\epsilon_0 \alpha}{V} E_{mic},
$$

where α and *V* are the polarizability and volume of the oscillator, respectively. Substituting the expression for E_{mic} in Eq. (2) (2) (2) , one has the second important equation

$$
P = \frac{xq}{V} + \frac{\epsilon_0 \alpha}{V} E + \frac{\eta \alpha}{V} P
$$

or

$$
P = \frac{1}{1 - \eta \alpha/V} \left(\frac{xq}{V} + \frac{\epsilon_0 \alpha}{V} E \right). \tag{4}
$$

Taken together, Eqs. (3) (3) (3) and (4) (4) (4) are enough for dealing with single-crystal problems (where there is translational symmetry). Next we discuss the dielectric response.

To obtain the expression for the dielectric constant which relates P and E , we must eliminate P from Eq. ([3](#page-0-2)). Plugging Eq. (4) (4) (4) into Eq. (3) (3) (3) , one has

$$
m\ddot{x} + m\gamma \dot{x} + m\omega_0^2 x = q \left[E + \frac{\eta}{\epsilon_0} \frac{1}{1 - \eta \alpha/V} \left(\frac{xq}{V} + \frac{\epsilon_0 \alpha}{V} E \right) \right],
$$

or

$$
m\ddot{x} + m\gamma\dot{x} + \left[m\omega_0^2 - \frac{q^2\eta}{\epsilon_0 V(1 - \eta\alpha/V)}\right]x = \frac{qE}{1 - \eta\alpha/V}.
$$

For $x = x_0 e^{-i\omega t}$, the solution is

$$
x = \frac{\frac{q}{m(1 - \eta \alpha/V)}E}{\left[\omega_0^2 - \frac{q^2 \eta}{m\epsilon_0 V(1 - \eta \alpha/V)}\right] - \omega^2 - i\gamma \omega}.
$$
 (5)

For simplicity, we define

$$
\omega_1^2 \equiv \frac{q^2 \eta}{m \epsilon_0 V (1 - \eta \alpha/V)}.
$$

Using Eq. ([4](#page-1-0)), we can write down the polarization

$$
P = \frac{1}{1 - \eta \alpha/V} \left(\frac{xq}{V} + \frac{\epsilon_0 \alpha}{V} E \right)
$$

=
$$
\frac{1}{1 - \eta \alpha/V} \left[\frac{q}{V} \frac{\frac{q}{m(1 - \eta \alpha/V)} E}{\omega_0^2 - \omega_1^2 - \omega^2 - i\gamma \omega} + \frac{\epsilon_0 \alpha}{V} E \right] = \frac{\epsilon_0 \alpha/V}{1 - \eta \alpha/V} E
$$

+
$$
\frac{q^2}{mV(1 - \eta \alpha/V)^2} \frac{E}{\omega_0^2 - \omega_1^2 - \omega^2 - i\gamma \omega}.
$$
 (6)

Therefore, the dielectric constant is

$$
\varepsilon = 1 + \frac{P}{\epsilon_0 E} = 1 + \frac{\alpha/V}{1 - \eta \alpha/V}
$$

$$
+ \frac{q^2}{\epsilon_0 m V (1 - \eta \alpha/V)^2} \frac{1}{\omega_0^2 - \omega_1^2 - \omega^2 - i\gamma \omega}.
$$
(7)

We can immediately see that the Lorentz model in a solid is modified, compared with that of an isolated oscillator due to the depolarization effect and the polarizability (when $\alpha = 0$ and $\eta = 0$, Eq. ([7](#page-1-1)) reduces to the case of an isolated oscillator).^{[19](#page-7-5)} We end this section by summarizing several useful definitions and expressions that connect measurable quantities (left-hand side) to microscopic parameters (righthand side). These include the high-frequency dielectric constant, oscillator strength, and TO phonon frequency.

$$
\varepsilon(\infty) \equiv 1 + \frac{\alpha/V}{1 - \eta \alpha/V},\tag{8a}
$$

$$
A = \frac{q^2}{\epsilon_0 m V (1 - \eta \alpha / V)^2},\tag{8b}
$$

$$
\omega_{\rm TO}^2 \equiv \omega_0^2 - \omega_1^2 = \omega_0^2 - \frac{q^2 \eta}{\epsilon_0 m V (1 - \eta \alpha / V)}.
$$
 (8c)

Clearly, one can extract $\varepsilon(\infty)$, A, and ω_{TO}^2 from the experimental vibrational spectra using an oscillator fit or Kramers Kronig analysis and use these quantities to extract the microscopic parameters.¹⁹ In other words, once $\varepsilon(\infty)$ and *A* are measured, one can derive α and q using Eqs. ([8a](#page-1-2)) and ([8b](#page-1-3)). Note that

$$
S \equiv \frac{A}{\omega_{\rm TO}^2}
$$

is the dimensionless oscillator strength.

B. Evaluating the effective charges

Ionic effective charge (also called local effective charge) *q* is an important quantity because it quantifies the ionicity. It is distinct from the Born effective charge as discussed below. The most straightforward way to extract ionic effective charge from oscillator strength is to use Eq. $(8b)$ $(8b)$ $(8b)$

$$
q^2 = \epsilon_0 m V (1 - \eta \alpha / V)^2 A. \tag{9}
$$

Thus, quantitative information about bond covalency/ionicity can be extracted from knowledge of oscillator strength and polarizability.

In the literature, we often find Born effective charge defined $as²¹$

$$
\frac{q_B^2}{\epsilon_0 m V} = \varepsilon(\infty) (\omega_{\text{LO}}^2 - \omega_{\text{TO}}^2),\tag{10}
$$

which, as shown in the Appendix, is equivalent to

$$
\frac{q_B^2}{\epsilon_0 m V} = A = \frac{q^2}{\epsilon_0 m V (1 - \eta \alpha / V)^2}.
$$
 (11)

Therefore Born and ionic effective charges are related as

$$
q_B^2 = \frac{q^2}{(1 - \eta \alpha/V)^2}.
$$
 (12)

Note that Born effective charge takes into account the combined contributions of ionic displacement, electron-cloud deformation, and depolarization effects, whereas ionic effective charge only represents the charge of the ions.

III. MULTIPLE COLLINEAR OSCILLATORS

When there is more than one oscillator (as in most real materials), one has to go back to the polarizability $[Eq. (6)]$ $[Eq. (6)]$ $[Eq. (6)]$ and add a mode index *j*. This yields

$$
P_j = \frac{\alpha_j/V}{1 - \eta_j \alpha_j/V} E + \frac{q_j^2}{m_j V (1 - \eta_j \alpha_j/V)^2} \frac{E}{\omega_0^2 - \omega_{1,j}^2 - \omega^2 - i\gamma_j \omega}
$$
\n(13)

and

$$
\omega_{1,j}^2 \equiv \frac{q_j^2 \eta_j}{\epsilon_0 m_j V (1 - \eta_j \alpha_j / V)}.
$$

Then the dielectric constant and other definitions can be expanded as

$$
\sum_{\epsilon=1}^{D} P_j
$$
\n
$$
\varepsilon = 1 + \frac{j}{\epsilon_0 E} = 1 + \sum_{j} \frac{\alpha_j / V}{1 - \eta_j \alpha_j / V}
$$
\n
$$
+ \sum_{j} \frac{q_j^2}{\epsilon_0 m_j V (1 - \eta_j \alpha_j / V)^2} \frac{1}{(\omega_0^2 - \omega_{1,j}^2) - \omega^2 - i \gamma_j \omega} = 1
$$
\n
$$
+ \sum_{j} [\varepsilon_j(\infty) - 1] + \sum_{j} A_j \frac{1}{\omega_{\text{TO},j}^2 - \omega^2 - i \gamma_j \omega}, \qquad (14)
$$

where

$$
\varepsilon_j(\infty) = 1 + \frac{\alpha_j/V}{1 - \eta_j \alpha_j/V},
$$
\n(15a)

$$
A_j = \frac{q_j^2}{\epsilon_0 m_j V (1 - \eta_j \alpha_j / V)^2},\tag{15b}
$$

$$
\omega_{\text{TO},j}^2 = \omega_{0,j}^2 - \omega_{1,j}^2 = \omega_{0,j}^2 - \frac{q_j^2 \eta_j}{\epsilon_0 m_j V (1 - \eta_j \alpha_j / V)}.
$$
 (15c)

Note that polarizability represents the high-frequency dielectric response of the electron cloud. Therefore, α should be labeled according to the polarization direction, although for simplicity, we label this quantity with the mode index.

IV. TILTED OSCILLATOR

If the electric field is not perfectly aligned with the direction of motion of a certain mode, the observed oscillator strength will be reduced from its intrinsic value. This can be easily understood by considering the extreme case: when the light is polarized perpendicular to a particular mode, this mode will not contribute to the oscillator strength at all.

Thus, if one directly employs the formulas for local ionic and Born effective charges in Sec. [II B](#page-1-5) without modification, the results will be unphysical. This is because a tilted oscillator provides only a component of the total intrinsic oscillator strength. Instead, we must go back to Sec. [II A](#page-0-4) and rederive a set of formulas that take orientation into account.

We can employ a modified version of Eq. (5) (5) (5) to account for the effect of a tilted oscillator

$$
x(\theta) = \frac{\frac{q}{m(1 - \eta \alpha/V)} E \cos(\theta)}{\omega_0^2 - \omega_1^2 - \omega^2 - i\gamma \omega}.
$$
 (16)

Hence,

$$
P(\theta) = \frac{1}{1 - \eta \alpha/V} \left[\frac{xq}{V} + \frac{\epsilon_0 \alpha}{V} E \cos(\theta) \right] = \frac{\epsilon_0 \alpha/V}{1 - \eta \alpha/V} E \cos(\theta)
$$

$$
+ \frac{q^2}{mV(1 - \eta \alpha/V)^2} \frac{E \cos(\theta)}{\omega_0^2 - \omega_1^2 - \omega^2 - i\gamma \omega}, \tag{17}
$$

and

$$
\varepsilon(\theta) = 1 + \frac{P(\theta)\cos(\theta)}{\epsilon_0 E} = 1 + \frac{\alpha/V \cos^2(\theta)}{1 - \eta \alpha/V} + \frac{q^2}{\epsilon_0 m V (1 - \eta \alpha/V)^2} \frac{\cos^2(\theta)}{\omega_0^2 - \omega_1^2 - \omega^2 - i\gamma \omega}.
$$
 (18)

Here, θ is the angle between the electric field and the direction in which oscillator intensity is maximum. If a measurement is done on a sample with a distribution of orientations $({\theta})$, the result is appropriately averaged as

$$
\tilde{\varepsilon} = \langle \varepsilon(\theta) \rangle = 1 + \frac{\alpha/V \langle \cos^2(\theta) \rangle}{1 - \eta \alpha/V} \n+ \frac{q^2}{\epsilon_0 m V (1 - \eta \alpha/V)^2} \frac{\langle \cos^2(\theta) \rangle}{\omega_0^2 - \omega_1^2 - \omega^2 - i \gamma \omega}.
$$
\n(19)

In this case, the observed parameters are related to the microscopic parameters as

$$
\widetilde{\varepsilon}(\infty) = 1 + \frac{\alpha/V \langle \cos^2(\theta) \rangle}{1 - \eta \alpha/V},
$$
\n(20a)

$$
\widetilde{A} = \frac{q^2}{\epsilon_0 m V (1 - \eta \alpha / V)^2} \langle \cos^2(\theta) \rangle, \tag{20b}
$$

$$
\tilde{\omega}_{\rm TO}^2 = \omega_0^2 - \omega_1^2 = \omega_0^2 - \frac{q^2 \eta}{\epsilon_0 m V (1 - \eta \alpha / V)}.
$$
 (20c)

Here, the brackets $\langle \rangle$ indicate directional averaging.

V. MULTIPLE TILTED OSCILLATORS

Most isotropic samples of real materials have several vibrational modes. We can write down the dielectric constant for the case of multiple tilted oscillators by combining Eqs. (13) (13) (13) and (17) (17) (17)

$$
\varepsilon({\theta_j}) = 1 + \sum_j \frac{P(\theta_j) \cos(\theta_j)}{\epsilon_0 E} = 1 + \sum_j \frac{\alpha_j / V}{1 - \eta_j \alpha_j / V} \cos^2(\theta_j)
$$

$$
+ \sum_j \frac{q_j^2 \cos^2(\theta_j)}{\epsilon_0 m_j V (1 - \eta_j \alpha_j / V)^2} \frac{1}{(\omega_0^2 - \omega_{1,j}^2) - \omega^2 - i \gamma_j \omega},
$$
(21)

and

$$
\varepsilon({\theta_j}) = 1 + \sum_j [\varepsilon_j(\infty) - 1] \cos^2(\theta_j) + \sum_j A_j \frac{\cos^2(\theta_j)}{\omega_{\text{TO}}^2 - \omega^2 - i\gamma \omega}.
$$
\n(22)

Note that the $\{\theta_i\}$ are related to an oscillator's polarization direction. Therefore, the number of independent θ_i may be less than the number of modes. Hence,

$$
\tilde{\varepsilon} = \langle \varepsilon(\{\theta_j\}) \rangle = 1 + \sum_j [\varepsilon_j(\infty) - 1] \langle \cos^2(\theta_j) \rangle \n+ \sum_j A_j \frac{\langle \cos^2(\theta_j) \rangle}{\omega_{\text{TO},j}^2 - \omega^2 - i\gamma_j \omega} = \tilde{\varepsilon}(\infty) \n+ \sum_j \widetilde{A}_j \frac{1}{\omega_{\text{TO},j}^2 - \omega^2 - i\gamma_j \omega},
$$
\n(23)

where the observed (apparent) parameters are

$$
\tilde{\varepsilon}(\infty) = 1 + \sum_{j} \frac{\alpha_j / V \langle \cos^2(\theta_j) \rangle}{1 - \eta_j \alpha_j / V} = 1 + \sum_{j} \left[\varepsilon_j(\infty) - 1 \right] \langle \cos^2(\theta_j) \rangle
$$
\n(24a)

$$
\widetilde{A_j} = \frac{q_j^2}{\epsilon_0 m_j V (1 - \eta_j \alpha_j / V)^2} \langle \cos^2(\theta_j) \rangle
$$
 (24b)

$$
\tilde{\omega}_{\text{TO},j}^2 = \omega_{0,j}^2 - \omega_{1,j}^2 = \omega_{0,j}^2 - \frac{q_j^2 \eta_j}{\epsilon_0 m_j V (1 - \eta_j \alpha_j / V)}.
$$
 (24c)

If all vibrational features can be resolved in frequency If all vibrational features can be resolved in frequency
space, $\tilde{\varepsilon}(\infty)$ and \tilde{A}_j can be determined from a model oscillator fit or a Kramers Kronig analysis. At the same time, the η_i can be estimated from the crystal structure. 20 However, one cannot use the $N+1$ equations given by Eqs. $(24a)$ $(24a)$ $(24a)$ and $(24b)$ $(24b)$ $(24b)$ to solve for 3*N* unknowns. The latter includes α_i , q_i , and $\langle \cos^2(\theta_j) \rangle$. Even if in some cases, we know $\langle \cos^2(\theta_j) \rangle$ (perhaps from an independent x-ray measurement), there are still 2*N* unknowns to determine from only *N*+ 1 equations. Additional information is needed to constrain the system.

Despite this limitation, vibrational spectroscopy of unoriented powdered samples can still be an important tool for extracting microscopic charge and bonding information. There are two important cases.

Case 1: The system is simple enough to have $N+1=2N$. For example, in rocksalts, the crystal symmetry is cubic $(\langle \cos^2(\theta_j) \rangle)$ is always 1/3), and there is only one vibrational mode $(N=1)$. Thus powder spectroscopy is sufficient to determine all of the microscopic parameters for systems such as NaCl or MnO.

Case 2: Occasionally, some variables are already known, say, from another method, sample, or similar compound, so that the total number of known variables can be reduced to be equal to or less than $N+1$. Recent work on MoS₂ nanoparticles provides a good example.¹⁶ Here, the interplane oscillator strength is identical to that of the single crystal. Therefore, it is reasonable to assume that the corresponding polarizability and charge are both the same for the nanoparticles as they are in the single crystal, a coincidence that reduces the number of unknowns and makes the extraction of intraplane charge bonding information possible. We elaborate on the case of $MoS₂$ below.

VI. DYNAMICS OF A MODEL TRANSITION METAL DICHALCOGENIDE: TESTING OUR APPROACH

In order to test the workability of this approach, we elected to investigate a model transition-metal dichalcogenide. The bulk material, $2H-MoS₂$, belongs to the $P6_3/mmc$ space group [Fig. [1](#page-4-0)(a)].^{[22](#page-7-8)} One consequence of this layered architecture is the low-dimensional electronic structure which consists of strong bonding within layers and weak van der Waals interactions between layers.²³ Each MoS₂ slab contains a layer of metal centers, sandwiched between two chalcogen layers, with each metal atom bonded to six chalcogen atoms in a trigonal prismatic arrangement. There are two infrared active E_{1u} and A_{2u} vibrational modes.²³ Schematic views of these displacement patterns are shown in Fig. [1](#page-4-0)(a). The E_{1u} and A_{2u} symmetries characterize intralayer and interlayer motions, respectively. We begin with demonstrating the self-consistency of the theory by analyzing the oscillator orientation and predicting the observed optical parameters. We then extend our technique to the chemically identical but morphologically different nanoparticles to illustrate the consequences of finite size, strain, and curvature.

Figure $1(b)$ $1(b)$ displays a close-up view of the reflectance of $2H-MoS_2$ ^{[24](#page-7-10)} As expected for a pressed powder sample, both the E_{1u} and A_{2u} modes are clearly observed. At normal incidence, the dielectric function is related to reflectance as

$$
R(\omega) = \left| \frac{\sqrt{\epsilon(\omega)} - 1}{\sqrt{\epsilon(\omega)} + 1} \right|^2.
$$
 (25)

Equation (25) (25) (25) is formally valid for single-bounce reflectance at the interface of two semi-infinite media. For the case of real materials with finite thickness, sample thickness must be sufficient to assume that there is no back reflectance. Large attenuation due to a strong resonance is helpful here.

A fit to the $2H-MoS_2$ powder data using Eqs. ([23](#page-3-4)) and ([25](#page-3-3)) [Fig. [1](#page-4-0)(b)] allows us to extract $\tilde{\varepsilon}(\infty) = 10.3$, $\tilde{S}_{E_{1u}} = 0.114$, and $\tilde{S}_{A_{2u}}$ = 0.0036 (Table [I](#page-4-1)). We refer to these values as "apparent oscillator parameters" because they are obtained from direct fits to the measured powder spectrum. They are distinct from the "intrinsic parameters" that are useful for evaluation of Born and ionic charges.

The apparent and intrinsic oscillator parameters are related according to Eqs. $(24a)$ $(24a)$ $(24a)$ – $(24c)$ $(24c)$ $(24c)$ by the oscillator orientation. For $MoS₂$, one has

FIG. 1. (Color online) (a) Crystal structure of 2H-MoS₂,Ref. [22,](#page-7-8) schematic view of the displacement patterns of the infrared-active modes, photo of a typical pressed pellet sample, and diagram for tilted oscillators. (b) Close-up view of the 300 K reflectance spectra of 2H-MoS₂ powder. The (red) solid curve is experimental data and the (blue) dashed line is a fit according to Eqs. (25) (25) (25) . The insets show the dielectric constant and a schematic view of the $2H-MoS₂$ platelets in a pressed pellet sample. (c) Close-up view of the 300 K reflectance of IF-MoS₂. An identical color scheme and line type are employed. The insets show the dielectric response and a high resolution TEM image of an IF-MoS₂ nanoparticle.

$$
\begin{aligned} \widetilde{\varepsilon}(\infty) &= 1 + \big[\varepsilon_{E_{1u}}(\infty) - 1\big] \langle \cos^2(\theta_{E_{1u}}) \rangle v + \big[\varepsilon_{A_{2u}}(\infty) - 1\big] \\ &\times \langle \cos^2(\theta_{A_{2u}}) \rangle v \,, \end{aligned} \tag{26a}
$$

TABLE I. Apparent parameters extracted from an oscillator fit to the measured reflectance spectrum of $2H-MoS₂$ powder, and the intrinsic parameters from a single-crystal sample in Ref. [17.](#page-7-12)

	\tilde{S}_i _a	$\tilde{\epsilon}(\infty)$ ^a	$S_i^{\ b}$	$\epsilon(\infty)$ b
E_{1u}	0.114	10.3	0.20	15.2
A_{2u}	0.0036		0.030	6.2

a Apparent parameters obtained from an oscillator fitting analysis of the measured powder spectrum.

^bIntrinsic parameters obtained from a fit to the measured reflectance of a single-crystal sample in Ref. [17.](#page-7-12)

$$
\widetilde{S}_{E_{1u}} = S_{E_{1u}} \langle \cos^2(\theta_{E_{1u}}) \rangle v, \tag{26b}
$$

$$
\widetilde{S}_{A_{2u}} = S_{A_{2u}} \langle \cos^2(\theta_{A_{2u}}) \rangle v. \tag{26c}
$$

Here, $v = 0.7$ is the relative density of the unoriented pressed pellet compared with that of the single crystal. $24,25$ $24,25$ A benchmark is now needed to relate the apparent and intrinsic oscillator parameters.

Fortunately, the polarized infrared reflectance of a $2H-MoS₂$ single crystal has been studied by Wieting and Verble.¹⁷ Fits to the reflectance yield $\epsilon(\infty)_{E_{1u}} = 15.2$, $\epsilon(\infty)_{A_{2u}}$ $= 6.2$, $S_{E_{1u}} = 0.20$, and $S_{A_{2u}} = 0.03$ (Table [I](#page-4-1)). These intrinsic parameters are appropriate benchmarks for our pellet samples, because they are made of μ m-sized powders, for which the surface effect can be ignored as a good approximation.

In Eqs. ([26a](#page-4-2))–([26c](#page-4-3)), the only unknowns are $\langle \cos^2(\theta_{E_{1u}}) \rangle$ and $\langle \cos^2(\theta_{A_{2u}}) \rangle$. Figure [1](#page-4-0)(b) shows a schematic view of the pressed pellet of 2H powder. Since the polarizations of the two modes are orthogonal, one has $\theta_{E_{1u}} + \theta_{A_{2u}} = \frac{\pi}{2}$. Hence, $\langle \cos^2(\theta_{E_{1u}}) \rangle + \langle \cos^2(\theta_{A_{2u}}) \rangle = 1$, meaning that there is only one unknown, say, $\langle \cos^2(\vec{\theta}_{E_{1u}}) \rangle$. Using Eqs. ([26b](#page-4-4)) and ([26c](#page-4-3)), we find two independent values as 0.81 and 0.83 in good agreement with each other. To further check the self-consistency, we calculated $\tilde{\epsilon}$ (∞) using Eq. ([26a](#page-4-2)) using an average value of $\langle \cos^2(\theta_{E_{1u}}) \rangle = 0.82$, yielding $\tilde{\epsilon}(\infty) = 9.8$, in excellent agreement with that obtained by direct fitting techniques $[\tilde{\epsilon}(\infty)]$ $= 10.3$. In addition to confirming the validity of our approach, this self-consistency also shows that the pressed powder sample of $2H-MoS₂$ will have the same Born effective charge as the single crystal, $16,17$ $16,17$ which, of course, it must. The dielectric response ε_1 was calculated using the intrinsic parameters of Ref. [16](#page-7-3) and is plotted in the inset of Fig. [1](#page-4-0)(b). The dispersive response is typical of an anisotropic insulator with two vibrational modes.

The availability of \sim 30–70 nm average diameter nested $MoS₂$ nanoparticles provides an opportunity to investigate the impact of finite length scale effects on chemical bonding.^{[1](#page-4-0)6} Figure 1(c) displays a close-up view of the measured far-infrared reflectance along with an oscillator fit. The apparent parameters obtained from this fit can be scaled toward a set of intrinsic oscillator parameters using the orientation and density corrections outlined above.¹⁶ From an analysis of these intrinsic oscillator strengths and highfrequency dielectric constants along with mode frequencies, we can extract Born and local effective charges for the nanoparticles. 16 In the intralayer direction, we find that the Born effective charge of the nanoparticles is 0.69*e* in the intralayer direction, significantly lower than that of the layered bulk $(1.11e)$. Here, *e* is the charge of an electron. We attribute this difference to structural strain (and resulting change in intralayer polarizability) in the nanoparticles.¹⁶ The Born effective charge of the nanoparticles remains unchanged in the interlayer direction $(0.52e)$. The dielectric constant was again calculated using intrinsic parameters of Ref. 16 and is plotted in the inset of Fig. $1(c)$ $1(c)$. Clearly, the dispersive response of the nanoparticles is different from that of the bulk in the intralayer direction. They are the same in the interlayer direction.

Extension of Born and local (or ionic) charges concepts to nanomaterials is an important advance because most are not (and may never be) available in an oriented form. $9-14$ $9-14$ Indeed, emerging mechanical and tribological applications of nanoscale $MoS₂$ require bulk quantities of powder with careful size-shape control but no orientational control. At the same time, the relationship between engineering properties such as solid-state lubrication^{26–[28](#page-7-15)} and the microscopic aspects of charge and bonding are an open and interesting question that deserves further study.

VII. CONCLUSION

We present an application of the Lorentz model in which fits to vibrational spectra or a Kramers Kronig analysis of the reflectance are employed along with several useful formalisms to quantify microscopic charge and polarizability in unoriented (powdered) materials. This paper provides a systematic development of the operative equations and a discussion of the conditions under which such techniques can be employed. We demonstrate the workability of our approach by analyzing the vibrational response of a layered transitionmetal dichalcogenide, and we include an evaluation of Born and local (or ionic charge) of its nanoscale analog to illustrate the modern utility. The extension to assess size effects advances the field of nanoscience and, at the same time, retains many attractive features of optical spectroscopy and the traditional Lorentz model.

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APPENDIX: OTHER FRAMEWORKS

For applications, it can be convenient to use other forms of Eq. (7) (7) (7) that are written in terms of parameters that are more related to the experimental observations. Before stepping into that, we take a careful look at Eq. (7) (7) (7) and note that there are only three independent parameters: α , q , and ω_0 . Other three variable sets offer equivalent expressions. Some of these are detailed below.

1. $\varepsilon(0)$, $\varepsilon(\infty)$, and ω_{TO}^2 frameworks

This is a very useful framework because $\varepsilon(0)$, $\varepsilon(\infty)$, and ω_{TO}^2 can all be straightforwardly extracted from the optical spectra.

a. Collinear oscillators

Equation $(8a)$ $(8a)$ $(8a)$ is already very close to employing this new set of parameters. Let us define

$$
\varepsilon(0) = \left(1 + \frac{\alpha/V}{1 - \eta \alpha/V}\right) + \frac{q^2}{\epsilon_0 m V (1 - \eta \alpha/V)^2} \frac{1}{\omega_{\text{TO}}^2} = \varepsilon(\infty)
$$

$$
+ \frac{q^2}{\epsilon_0 m V (1 - \eta \alpha/V)^2} \frac{1}{\omega_{\text{TO}}^2}.
$$
(A1)

Then,

$$
A = \frac{q^2}{mV(1 - \eta \alpha/V)^2} = \omega_{\text{TO}}^2[\varepsilon(0) - \varepsilon(\infty)]
$$

with parameters $\varepsilon(0)$, $\varepsilon(\infty)$, and ω_{TO}^2 , one can rewrite Eq. ([7](#page-1-1))

$$
\varepsilon = \varepsilon(\infty) + \omega_{\text{TO}}^2 \frac{\varepsilon(0) - \varepsilon(\infty)}{\omega_{\text{TO}}^2 - \omega^2 - i\gamma\omega}.\tag{A2}
$$

(i) ω_0^2

It is a bit tedious, but not so difficult to show that

$$
\omega_0^2 = \omega_{\text{TO}}^2 \frac{\varepsilon(\infty) + 1/\eta - 1}{\varepsilon(0) + 1/\eta - 1}.
$$

(ii) Ionic effective charge

$$
q^{2} = \epsilon_0 m V (1 - \eta \alpha/V)^{2} A = \epsilon_0 m V (1 - \eta \alpha/V)^{2} \omega_{\text{TO}}^{2} [\varepsilon(0) - \varepsilon(\infty)]
$$

$$
= \epsilon_0 m V \frac{\omega_{\text{TO}}^2[\varepsilon(0) - \varepsilon(\infty)]}{\eta^2[\varepsilon(\infty) + 1/\eta - 1]^2}.
$$
 (A3)

(iii) Born effective charge

$$
q_B^2 = \epsilon_0 m V \omega_{\text{TO}}^2 [\varepsilon(0) - \varepsilon(\infty)] \tag{A4}
$$

(iv) ω_{LO}^2

By definition, ω_{LO} , $\varepsilon = 0$ for a longitudinal mode. Using Eq. ([A2](#page-5-0)) (and ignoring $i\gamma\omega$), one has

$$
0 = \varepsilon(\infty) + \omega_{\text{TO}}^2 \frac{\varepsilon(0) - \varepsilon(\infty)}{\omega_{\text{TO}}^2 - \omega_{\text{LO}}^2}
$$
 (A5)

which gives the Lyddane-Sachs-Teller relation,³

$$
\frac{\omega_{\text{LO}}^2}{\omega_{\text{TO}}^2} = \frac{\varepsilon(0)}{\varepsilon(\infty)}\tag{A6}
$$

One can write out ω_{LO}^2 in terms of microscopic parameters α , *q*, and ω_0 using the Lyddane-Sachs-Teller relation as follows:

$$
\omega_{LO}^2 = \omega_{TO}^2 \frac{\varepsilon(0)}{\varepsilon(\infty)} = \omega_{TO}^2 \frac{\varepsilon(\infty) + A \frac{1}{\omega_{TO}^2}}{\varepsilon(\infty)} = \omega_{TO}^2 + \frac{A}{\varepsilon(\infty)} = \omega_0^2
$$

$$
- \frac{q^2 \eta}{\varepsilon_0 m V (1 - \eta \alpha/V)} + \frac{q^2}{\varepsilon_0 m V (1 - \eta \alpha/V)^2} \frac{1}{1 + \frac{\alpha/V}{1 - \eta \alpha/V}}
$$

$$
= \omega_0^2 + \frac{q^2}{\varepsilon_0 m V} \frac{1 - \eta}{1 + (1 - \eta) \alpha/V}.
$$
 (A7)

Note that ω_{LO}^2 is large than ω_0^2 , while ω_{TO}^2 is smaller than ω_0^2 .

b. Multiple collinear oscillators

$$
\varepsilon = 1 + \sum_{j} \left[\varepsilon_{j}(\infty) - 1 \right] + \sum_{j} \omega_{\text{TO},j}^{2} \frac{\varepsilon_{j}(0) - \varepsilon_{j}(\infty)}{\omega_{\text{TO},j}^{2} - \omega^{2} - i\gamma_{j}\omega}.
$$
\n(A8)

c. Tilted oscillators

$$
\varepsilon = 1 + [\varepsilon(\infty) - 1)]\langle \cos^2(\theta) \rangle + \omega_{\text{TO}}^2 \frac{\varepsilon(0) - \varepsilon(\infty)}{\omega_{\text{TO}}^2 - \omega^2 - i\gamma\omega} \langle \cos^2(\theta) \rangle. \tag{A9}
$$

d. Multiple tilted oscillators

$$
\varepsilon(\{\theta_j\}) = 1 + \sum_j [\varepsilon_j(\infty) - 1] \cos^2(\theta_j)
$$

$$
+ \sum_j \omega_{\text{TO},j}^2 \frac{\varepsilon_j(0) - \varepsilon_j(\infty)}{\omega_{\text{TO},j}^2 - \omega^2 - i\gamma_j \omega} \cos^2(\theta_j)
$$

$$
\tilde{\varepsilon} = \langle \varepsilon(\{\theta_j\}) \rangle = 1 + \sum_j [\varepsilon_j(\infty) - 1] \langle \cos^2(\theta_j) \rangle
$$

$$
+ \sum_j \omega_{\text{TO},j}^2 \frac{\varepsilon_j(0) - \varepsilon_j(\infty)}{\omega_{\text{TO},j}^2 - \omega^2 - i\gamma_j \omega} \langle \cos^2(\theta_j) \rangle \tag{A10}
$$

2. $\epsilon(\infty)$, ω_{TO}^2 , and ω_{LO}^2 frameworks

a. Collinear oscillators

Another choice is to use $\varepsilon(\infty)$, ω_{TO}^2 , and ω_{LO}^2 . Using the Lyddane-Sachs-Teller relation, Eq. ([A6](#page-5-1)), one can eliminate $\varepsilon(0)$, which yields

$$
\varepsilon = \varepsilon(\infty) + \varepsilon(\infty) \frac{\omega_{\text{LO}}^2 - \omega_{\text{TO}}^2}{\omega_{\text{TO}}^2 - \omega^2 - i\gamma\omega}.
$$
 (A11)

(i) Ionic effective charge

$$
q^{2} = \epsilon_{0} m V (1 - \eta \alpha/V)^{2} A = \epsilon_{0} m V (1 - \eta \alpha/V)^{2} \epsilon(\infty) (\omega_{LO}^{2} - \omega_{TO}^{2})
$$

$$
= \epsilon_{0} m V \frac{\epsilon(\infty) (\omega_{LO}^{2} - \omega_{TO}^{2})}{\eta^{2} [\epsilon(\infty) + 1/\eta - 1]^{2}}.
$$
(A12)

(ii) Born effective charge

$$
q_B^2 = \epsilon_0 m V \varepsilon(\infty) (\omega_{\text{LO}}^2 - \omega_{\text{TO}}^2). \tag{A13}
$$

b. Multiple collinear oscillators

$$
\varepsilon = 1 + \sum_{j} \left[\varepsilon_{j}(\infty) - 1 \right] + \sum_{j} \varepsilon_{j}(\infty) \frac{\omega_{\text{LO},j}^{2} - \omega_{\text{TO},j}^{2}}{\omega_{\text{TO},j}^{2} - \omega^{2} - i\gamma_{j}\omega}.
$$
\n(A14)

c. Tilted oscillators

$$
\varepsilon = 1 + [\varepsilon(\infty) - 1] \langle \cos^2(\theta) \rangle + \varepsilon(\infty) \frac{\omega_{\text{LO}}^2 - \omega_{\text{TO},j}^2}{\omega_{\text{TO}}^2 - \omega^2 - i\gamma \omega} \langle \cos^2(\theta) \rangle. \tag{A15}
$$

d. Multiple tilted oscillators

$$
\varepsilon({\theta_j}) = 1 + \sum_j [\varepsilon_j(\infty) - 1] \cos^2(\theta_j)
$$

+
$$
\sum_j \varepsilon_j(\infty) \frac{\omega_{\text{LO},j}^2 - \omega_{\text{TO},j}^2}{\omega_{\text{TO},j}^2 - \omega^2 - i\gamma_j \omega} \cos^2(\theta_j)
$$
 (A16)

$$
\tilde{\varepsilon} = \langle \varepsilon(\{\theta_j\}) \rangle = 1 + \sum_j [\varepsilon_j(\infty) - 1] \langle \cos^2(\theta_j) \rangle
$$

+
$$
\sum_j \varepsilon_j(\infty) \frac{(\omega_{1O,j}^2 - \omega_{\text{TO},j}^2)}{\omega_{\text{TO},j}^2 - \omega^2 - i\gamma_j \omega} \langle \cos^2(\theta_j) \rangle. \quad (A17)
$$

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- ²³ J. L. Verble and T. J. Wieting, Phys. Rev. Lett. **25**, 362 (1970).
- 24 IF-MoS₂ was prepared as described previously in Ref. [16.](#page-7-3) $2H-MoS_2$ was purchased directly from Alfa Aesar (99%). Pressed pellet samples were prepared using low pressure [see Fig. [1](#page-4-0)(a)]. The theoretical density of $2H-MoS_2$ single crystal is 4.996 $g/cm³$, and the actual densities of 2H and IF pellets are \sim 3.5 g/cm³. Pellet densities are therefore \sim 70% of the singlecrystal density, a difference that we correct for in our analysis. We checked that there is no significant infrared transmittance of the samples (thickness between 1.0 and 1.3 mm), indicating large attenuation due to strong resonance. Near normal infrared reflectance was measured using a Bruker 113V Fourier transform infrared spectrometer. A helium-cooled bolometer detector was employed in the far-infrared for added sensitivity. Both 0.5 and 2 cm⁻¹ resolutions were used in the infrared (Ref. [16](#page-7-3)).
- ²⁵For far-infrared spectroscopy, the wavelength of the light is usually much larger than the particle size. Therefore, the measured oscillator strength is proportional to the concentration of the oscillators, which is inversely proportional to the mass density of the powder sample.
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